Table 3. Selected bond distances and angles in ammonium azide
Distances and angles marked with an asterisk are involved in hydrogen bonds.

|  | Uncorrected | Corrected |
| :--- | :---: | :---: |
| $\mathrm{N}(1)-\mathrm{N}(2)$ | $1.168(4) \AA$ | $1.186 \AA$ |
| $\mathrm{~N}(3)-\mathrm{N}(4)$ | $1.171(4)$ | 1.186 |
| $\mathrm{~N}(5)-\mathrm{H}(1)^{*}$ | $1.037(10)$ | 1.067 |
| $\mathrm{~N}(5)-\mathrm{H}(2)^{*}$ | $1.001(9)$ | 1.028 |
| $\mathrm{~N}(5)-\mathrm{N}(2)^{*}$ | $2.967(3)$ |  |
| $\mathrm{N}(5)-\mathrm{N}(4)^{*}$ | $2.975(4)$ |  |
| $\mathrm{N}(2)-\mathrm{H}(1)^{*}$ | $1.930(9)$ |  |
| $\mathrm{N}(4)-\mathrm{H}(2)^{*}$ | $1.74(9)$ |  |
| $\mathrm{H}\left(1-\mathrm{N}(5)-\mathrm{H}\left(1^{\prime}\right)\right.$ | $106.0(1.1)^{\circ}$ |  |
| $\mathrm{H}(1)-\mathrm{N}(5)-\mathrm{H}(2)$ | $110.3(0.7)$ |  |
| $\mathrm{H}(1)-\mathrm{N}(5)-\mathrm{H}\left(2^{\prime}\right)$ | $110.3(0.7)$ |  |
| $\mathrm{H}(2)-\mathrm{N}(5)-\mathrm{H}\left(2^{\prime}\right)$ | $109.6(1.0)$ |  |
| $\mathrm{N}(4)-\mathrm{N}(5)-\mathrm{N}(2)$ | $109.4(0.1)$ |  |
| $\mathrm{N}(4)-\mathrm{N}(5)-\mathrm{N}\left(2^{\prime}\right)$ | $108.5(0.1)$ |  |
| $\mathrm{N}(4)-\mathrm{N}(5)-\mathrm{N}\left(4^{\prime}\right)$ | $112.5(0.2)$ |  |
| $\mathrm{N}(2)-\mathrm{N}(5)-\mathrm{N}\left(2^{\prime}\right)$ | $108.4(0.1)$ |  |
| $\mathrm{N}(5)-\mathrm{H}(1)-\mathrm{N}(2)^{*}$ | $177.9(0.9)$ |  |
| $\mathrm{N}(5)-\mathrm{H}(2)-\mathrm{N}(4)^{*}$ | $177.2(0.7)$ |  |

bonds. The hydrogen bonds are nearly linear. The $\mathrm{N}-\mathrm{H}$ bond distances within the ammonium group differ by about four times the estimated standard deviation, which is probably significant.

Differences in thermal ellipsoids between constrained and unconstrained models were insignificant, except
that the shortest r.m.s. amplitudes for the hydrogen atoms, which were essentially parallel to the $\mathrm{N}-\mathrm{H}$ bonds in both models, were about $0.02 \AA$ shorter when the ammonium group was held rigid. This would seem to be a reasonable amplitude for an $\mathrm{N}-\mathrm{H}$ stretching vibration within a strong hydrogen bond.

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# Plutonium-Platinum $\mathbf{P u}_{\mathbf{2}} \mathbf{P t}^{*}$ 

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#### Abstract

Orthorhombic, Pnma, $Z=4, a=7.046$ (3), $b=4.707$ (2), $c=8.865$ (4) $\AA, \rho_{c}=15.20 \mathrm{~g} \mathrm{~cm}^{-3}$, $\mu=1115 \mathrm{~cm}^{-1}, \lambda=0.70930 \AA$. This structure belongs to the $C 23, \mathrm{PbCl}_{2}$ type. Introduction. Kutaitsev, Chebotarev, Lebedev, Andrianov, Konev \& Menshikova (1965) reported five compounds in the $\mathrm{Pu}-\mathrm{Pt}$ system: $\mathrm{Pu}_{5} \mathrm{Pt}_{3}, \mathrm{PuPt}, \mathrm{PuPt}_{2}$, $\mathrm{PuPt}_{3}$ and $\mathrm{PuPt}_{5}$. They reported that PuPt has the CrB , C33 structure, $\mathrm{PuPt}_{2}$ has the $\mathrm{MgCu}_{2}, C 15$ structure and that $\mathrm{PuPt}_{3}$ has the $\mathrm{AuCu}_{3}, L 1_{2}$ structure. Cromer \& Larson (1975) reported that $\mathrm{Pu}_{5} \mathrm{Pt}_{3}$ has the $\mathrm{Mn}_{5} \mathrm{Si}_{3}$, $D 8_{8}$ structure. * Work performed under the auspices of the Department of Energy.


The $\mathrm{Pu}-\mathrm{Pt}$ system is much more complex than Kutaitsev et al. (1965) found. Cromer \& Larson (1977) have since reported the existence and structure of $\mathrm{Pu}_{31} \mathrm{Pt}_{20}$. Land, Peterson \& Roof (1978) have reinvestigated this system, particularly the platinum-rich end. They verified the presence of $\mathrm{PuPt}_{5}$ and found that it has the $\mathrm{CaCu}_{5}, D 2_{d}$ structure. They also found the compound $\mathrm{PuPt}_{4}$ and proposed an orthorhombic structure based on a distortion of two unit cells of $\mathrm{PuPt}_{5}$ with the twofold Pt atoms removed. In addition, they found two high-temperature forms of $\mathrm{PuPt}_{3}$, one of which appears to be of the $\mathrm{Ni}_{3} \mathrm{P}, D 0_{e}$ type. The other $\mathrm{PuPt}_{3}$ phase is poorly characterized.

In the present note the existence and structure of yet another phase, $\mathrm{Pu}_{2} \mathrm{Pt}$, is reported.

An alloy of stoichiometry $\mathrm{Pu}_{2} \mathrm{Pt}$ was prepared by arc melting. It was subsequently heat treated at $550^{\circ} \mathrm{C}$ for about two months. During this period, because of a malfunction of the furnace control, the sample was inadvertently heated to $850^{\circ} \mathrm{C}$ for a short time. According to the phase diagram given by Kutaitsev et al. (1965), this $850^{\circ} \mathrm{C}$ temperature lies in the two-phase region of solid $\mathrm{Pu}_{5} \mathrm{Pt}_{3}(0.82$ mole fraction) in equilibrium with liquid of composition $15 \mathrm{at} . \% \mathrm{Pt}$. This phase diagram is, of course, incorrect in this region as no mention is made of $\mathrm{Pu}_{2} \mathrm{Pt}$. After cooling, a very small amount of metallic mirror was present on the inside of the quartz container, suggesting that a small amount of liquid had been present and had wet the container surface. Whatever the true nature of the phase diagram this liquid would have been richer in Pu and the remaining, solid ingot would have been slightly richer in Pt than the intended $\mathrm{Pu}_{2} \mathrm{Pt}$.

A portion of the ingot was crushed and searched for single crystals. A single crystal was soon found and precession photographs showed it to be orthorhombic, Pnma, if centrosymmetric. Reflections $0 k l$ were present only if $k+l=2 n$, and reflections $h k 0$ were present only if $h=2 n$. The structure determination subsequently confirmed Pnma.

Lattice constants were obtained from a least-squares refinement of 12 high-angle reflections measured on an automated Picker diffractometer. Intensity data were collected and processed in the manner described by Cromer \& Larson (1977). A complete sphere of reflections was measured. Crystallographic data are given in the Abstract and details of data collection are given in Table 1. Empirical absorption corrections were based on the variation of intensities as a function of $\varphi$ for a few reflections near $\chi=90^{\circ}$ (Furnas, 1957). In addition, spherical absorption corrections were applied based on the estimated mean radius of the irregularly shaped crystal.

Table 1. Data collection and reduction (Cromer \& Larson, 1977)

| Radiation | Mo Ka (graphite monochromated) |
| :--- | :---: |
| $2 \theta_{\text {max }}$ | $50^{\circ}$ |
| Take-off angle | $3{ }^{\circ}$ |
| Step size | $0.05^{\circ} 2 \theta$ |
| Step time | 2 s |
| Scan range | $2^{\circ}\left(+\alpha_{1}-\alpha_{2}\right.$ separation) |
| $T_{\text {min }}$ (threshold parameter) | 2.0 |
| Number of reflections | 1918 |
| Unique reflections | 294 |
| Unique reflections observed | 281 |
| Reflections measured twice | 11 |
| Reflections measured four times | 92 |
| Reflections measured eight times | 191 |
| $R_{F}$ (no corrections) | 0.146 |
| $R_{F}$ (empirical corrections) | 0.076 |
| Range of corrective factors | 0.020 to 0.138 |
|  |  |

The space group and axial ratios suggested that the structure might be of the $\mathrm{C} 23, \mathrm{PbCl}_{2}$ type. A structurefactor calculation using the parameters of $\mathrm{Ca}_{2} \mathrm{Si}$ gave good qualitative agreement with the precession photographs. A full-matrix least-squares refinement which minimized $\sum w(\Delta F)^{2}$ was made with anisotropic thermal parameters.* Scattering factors were taken from Cromer \& Waber (1974) and anomalous dispersion corrections from Cromer \& Liberman (1970). Positional parameters are given in Table 2 along with equivalent isotropic thermal parameters.

The stoichiometry was inferred from the structure type. No attempt was made to include occupancy or composition parameters as these two atoms are too close in scattering power to make such a refinement very meaningful.

Discussion. A number of intermetallic compounds have the $\mathrm{PbCl}_{2}$ structure type, but only about $14 \%$ of them (Pearson, 1967) contain only truly metallic elements. Most of the compounds of this type contain the elements $\mathrm{Si}, \mathrm{Ge}, \mathrm{Se}, \mathrm{P}, \mathrm{As}$ or H . Aside from the variable axial ratios of the orthorhombic cell, each of the atoms has variable positional parameters. Consequently the

[^0]Table 2. Final least-squares parameters for $\mathrm{Pu}_{2} \mathrm{Pt}$

|  | $x$ | $y$ | $z$ | Equivalent <br> isotropic $B$ |
| :--- | :---: | :---: | :---: | :---: |
|  | $x$ | $y(1)$ | $0.8472(2)$ | $\frac{1}{4}$ |
| Pu | $0.0795(1)$ | $1.35 \AA^{2}$ |  |  |
| $\mathrm{Pu}(2)$ | $0.9842(2)$ | $\frac{4}{4}$ | $0.6781(1)$ | 1.12 |
| Pt | $0.2589(2)$ | $\frac{1}{4}$ | $0.1002(1)$ | 1.49 |

Table 3. Interatomic distances $(\AA)$ in $\mathrm{Pu}_{2} \mathrm{Pt}$

| $\mathrm{Pu}(1)-\mathrm{Pt}$ | $2.907(2)$ | $\mathrm{Pu}(2)-\mathrm{Pt}$ | $2.934(2)$ |
| :--- | :--- | :--- | :--- |
| $\mathrm{Pu}(1)-\mathrm{Pt}$ | $2.907(3)$ | $\mathrm{Pu}(2)-2 \mathrm{Pt}$ | $3.048(2)$ |
| $\mathrm{Pu}(1)-2 \mathrm{Pt}$ | $2.938(1)$ | $\mathrm{Pu}(2)-2 \mathrm{Pt}$ | $3.512(2)$ |
| $\mathrm{Pu}(1)-\mathrm{Pt}$ | $4.149(3)$ | $\mathrm{Pu}(2)-\mathrm{Pt}$ | $4.212(2)$ |
| $\mathrm{Pu}(1)-2 \mathrm{Pt}(1)$ | $3.487(1)$ | $\mathrm{Pu}(2)-2 \mathrm{Pu}(1)$ | $3.01(2)$ |
| $\mathrm{Pu}(1)-2 \mathrm{Pu}(2)$ | $3.401(2)$ | $\mathrm{Pu}(2)-2 \mathrm{Pu}(1)$ | $3.429(2)$ |
| $\mathrm{Pu}(1)-2 \mathrm{Pu}(2)$ | $3.429(2)$ | $\mathrm{Pu}(2)-\mathrm{Pu}(1)$ | $3.429(2)$ |
| $\mathrm{Pu}(1)-\mathrm{Pu}(2)$ | $3.429(2)$ | $\mathrm{Pu}(2)-\mathrm{Pu}(1)$ | $3.687(2)$ |
| $\mathrm{Pu}(1)-\mathrm{Pu}(2)$ | $3.687(2)$ | $\mathrm{Pu}(2)-2 \mathrm{Pu}(2)$ | $3.746(1)$ |
| $\mathrm{Pt}-\mathrm{Pu}(1)$ |  | $2.907(2)$ | $\mathrm{Pu}(2)-2 \mathrm{Pu}(2)$ |
| P | $3.945(1)$ |  |  |
| $\mathrm{Pt}-\mathrm{Pu}(1)$ | $2.907(3)$ |  |  |
| $\mathrm{Pt}-2 \mathrm{Pu}(1)$ | $2.938(1)$ |  |  |
| $\mathrm{Pt}-\mathrm{Pu}(1)$ | $4.149(3)$ |  |  |
| $\mathrm{Pt}-\mathrm{Pu}(2)$ | $2.934(2)$ |  |  |
| $\mathrm{Pt}-2 \mathrm{Pu}(2)$ | $3.048(2)$ |  |  |
| $\mathrm{P}-2 \mathrm{Pu}(2)$ | $3.512(2)$ |  |  |
| $\mathrm{Pt}-\mathrm{Pu}(2)$ | $4.212(2)$ |  |  |



Fig. 1. The $\mathrm{Pu}(1)$ and Pt polyhedra in $\mathrm{Pu}_{2} \mathrm{Pt}$. View is approximately along $x$. The origin is at the lower left rear. $\mathrm{Pu}(1)$ is at $0.847, \frac{1}{4}, 0.079$ and the Pt is at $0.741, \frac{3}{4}, 0.900$.
polyhedra and the number of neighbors for each of the atoms can vary considerably for compounds nominally of the same structure type.

The neighbors and polyhedra were determined as described by Cromer \& Larson (1977). The resulting interatomic distances are given in Table 3. These distances are similar to those found in other $\mathrm{Pu}-\mathrm{Pt}$ compounds. $\mathrm{Pu}(1)$ has five Pt and eight Pu neighbors. There are five neighbors in the mirror plane and four on either side of this plane. $\mathrm{Pu}(2)$ has six Pt neighbors and ten Pu neighbors with six neighbors in the mirror plane and five on either side. The Pt atom has 11 neighbors, all Pu . There are five neighbors in the mirror plane and three on either side. Stereo drawings of these polyhedra are shown in Figs. 1 and 2.

Thanks are due to R. E. Tate for preparation of the alloy. All calculations were done on a CDC-7600 computer using the $L A S L$ system of crystallographic programs developed primarily by A. C. Larson.


Fig. 2. The $\mathrm{Pu}(2)$ polyhedron in $\mathrm{Pu}_{2} \mathrm{Pt} . \mathrm{Pu}(2)$ is at $0.984, \frac{1}{4}, 0.678$.

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# Oxysulfure de Gallium et de Lanthane $\mathbf{L a G a O S}_{\mathbf{2}}$ 

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Abstract. Orthorhombic, space group $P m c a, a=\quad \mu=203 \mathrm{~cm}^{-1}$. The structure was solved from single-
$5 \cdot 5419(9), b=5 \cdot 767(1), c=11.443(1) \AA, Z=4 . \quad \begin{aligned} & \text { crystal data. With anisotropic thermal parameters, the }\end{aligned}$


[^0]:    * Listings of the final least-squares cycle containing the anisotropic thermal parameters, structure factors, thermal ellipsoid calculations, interatomic distance calculations and polyhedron calculations have been deposited with the British Library Lending Division as Supplementary Publication No. SUP 33532 ( 15 pp .). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

